

**METAL CATALYSED INTUMESCENCE OF  
POLYHYDROXYL COMPOUNDS**

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# METAL CATALYSED INTUMESCENCE OF POLYHYDROXYL COMPOUNDS

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# METAL CATALYSED INTUMESCENCE OF POLYHYDROXYL COMPOUNDS

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## SYNOPSIS

Fire presents a great risk to humans and their possessions. Polymeric compounds are highly flammable and compounds are added to increase their fire resistance. These additives are referred to as flame retardants. Intumescence flame retardants are systems that form a carbon rich foam on the surface of the polymer during thermal degradation. A new field of metal catalysed intumescence is discussed in this document. The influence of both the metal ion and organic backbone of polyhydroxylcarboxylic acid metal salts was studied. The intumescence and foaming properties of selected salts were studied in more detail.

Gluconic acid proved to be the best intumescence and showed the best combination of carbon char yield, foam structure and volume, stability and open flame characteristics. As a general rule of thumb, the char volume and structure improved with an increased number of hydroxyl groups in the compound. Carbon char yield increased with an increase in the number of hydroxyl groups for a constant number of carbons in the complex. The number of acid groups in the compound played a lesser role in intumescence. Most metal complexes catalyse intumescence decomposition of polyhydric compounds such as polyhydroxylcarboxylic acids and pentaerythritol. The reduction in the volatilisation losses implies that the metal cations catalyse carbonisation reactions in the polyhydroxyl compound. Unfortunately, the resultant carbon-foams are unstable: the metal residues also catalyse the further oxidative degradation of the char. The low thermal stability of the compositions tested, as well as the afterglow effect makes them unsuitable for use as flame retardant additives in plastics. From the studies of the different metal complexes with acetylacetone and gluconic acid, it was concluded that the calcium ion exhibited the most promise.

When pyrolysed in air with an open gas flame or in a furnace, calcium gluconate monohydrate produces voluminous foam. The foam is of a closed cell structure, densely packed and has no mechanical strength. The cell walls are very thin (5 to 50 nm) and the cells are between 50 µm and 200 µm in size. If the sample is compressed before heating, the resultant foam produced during pyrolysis is less voluminous than that of the loose powder. The BET surface area of the calcium gluconate monohydrate foamed at 300°C for 5 min. is 16.0 m<sup>2</sup>/g. The foam produced after 5 min heating at 300°C has a thermal conductivity similar to standard polystyrene foam.

Adding leached silica to the calcium gluconate monohydrate increases mechanical strength of the foam, but influences the foam volume negatively. The optimum silica level was calculated to be 1:1 gluconate to silica on a mole ratio (11.8% SiO<sub>2</sub> by mass). The mechanical strength of the foam can also be increased with the addition of expandable graphite. The graphite has a lesser influence on the foam volume than the silica, but also reduces the foam volume. However, the addition of the expandable graphite gives the sample more “opening” force, as the foam volume of a compressed sample pyrolysed, is similar to that of the heated loose powder.

Calcium gluconate monohydrate starts to degrade at 120°C, losing its crystal water and shows a mass loss of 4% at 180°C. The exothermic peak (shown in the DSC/TGA data) associated with the metal catalysed carbon oxidation (afterglow) is observed at 570°C in air. For the calcium gluconate monohydrate the transition from CaCO<sub>3</sub> to CaO occurs above 650°C in air.

It has been shown that the foaming of polyhydroxylcarboxylic acids is due to the formation of water vapour during degradation. For calcium gluconate monohydrate, foaming starts due to the loss of crystal water and is continued by the loss of hydroxyl groups as water. The bulk of the foaming is due to the second reaction. It has been shown that compounds with crystal waters produce a more voluminous and lower density foam. The foam is an amorphous carbon rich residue. The molecular mass of the carbon residue increases up to a heating temperature of 300°C. This implies that the carbon residue crosslinks during formation, forming a stretchable cell wall for the foam. Compounds with a “free” hydroxyl group at the end of the carbon chain produce a foam of larger volume and lower density. This supports the crosslinking theory. The foam produced when calcium

gluconate monohydrate is heated for 5 min at 300°C in air is of very low density – 2.5 kg/m<sup>3</sup> based on residual mass.

The density of the calcium gluconate monohydrate pyrolysed at 1000°C for 5 min in air yields a CaO with a density of 20 kg/m<sup>3</sup>. This implies that the high temperature pyrolysis of calcium gluconate can produce an inorganic oxide of low bulk density and possible high specific surface area. The BET surface area of CaCO<sub>3</sub> from the gluconate pyrolysed at 600°C is ~ 12 m<sup>2</sup>/g.

An intumescent coating containing calcium gluconate monohydrate, leached silica and expandable graphite as a system was prepared and compared to commercial formulations. This gluconate based system was at least as efficient as the commercial formulations when painted on balsa wood planks or aluminium plates. On cardboard sheets it did not perform as well as the commercial systems.

More work should be done to overcome the afterglow effect observed with metal-based intumescent systems. The crystal structure of calcium gluconate monohydrate should be determined as to understand the decomposition better. It is unclear whether the decomposition is catalysed inter- or intramolecularly.

**KEY WORDS:** carbon foam, char, latent base catalysed intumescence, metal catalysed intumescence, carbonisation, intumescent flame retardants

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A PhD dissertation is never complete. There is always something else still to be done, something more to be added, to make it ... even more complete.

## CONTENT

<b>SYNOPSIS.....</b>	<i>i</i>
<b>KEY WORDS: .....</b>	<i>iii</i>
<b>ACKNOWLEDGEMENTS.....</b>	<i>iv</i>
<b>CONTENT.....</b>	<i>v</i>
<b>LIST OF FIGURES.....</b>	<i>x</i>
<b>LIST OF TABLES .....</b>	<i>xii</i>
<b>NOMENCLATURE .....</b>	<i>xiii</i>
<b>ABBREVIATIONS .....</b>	<i>xiii</i>
<b>1. Introduction.....</b>	<b>1-1</b>
<b>2. Background.....</b>	<b>2-1</b>
2.1. Thermal degradation, flammability and flame retardancy .....	2-1
2.2. The history of flame retardants.....	2-5
2.3. Flame retardants and synergism .....	2-7
2.4. Inorganic flame retardants .....	2-7
2.5. Halogenated flame retardants .....	2-10
2.6. Phosphorous flame retardants – Non intumescent.....	2-11
2.6.1. Condensed phase .....	2-12
2.6.2. Vapour phase flame retardants .....	2-13
2.7. Intumescent flame retardants.....	2-13
2.8. Oxygenated hydrocarbon flame retardants.....	2-18
2.9. Smoke suppressants.....	2-18
2.10. Metals in flame retardants .....	2-19
2.11. Fire testing.....	2-20
<b>3. Experimental.....</b>	<b>3-1</b>
3.1. Planning.....	3-1
3.1.1. The effect of the organic part of the organometallic complex .....	3-1
3.1.2. The effect of the metal cation of the organometallic complex .....	3-2

---

3.1.3.	The intumescence of calcium gluconate monohydrate and ammonium gluconate hydrate.....	3-7
3.1.4.	The foaming properties of metal glucose (dextrose) derivatives.....	3-9
3.1.5.	The formation of fine metal carbonates and oxides.....	3-11
3.2.	Apparatus.....	3-12
3.2.1.	Laboratory furnace .....	3-12
3.2.2.	Pyrolysis glass tubes.....	3-12
3.2.3.	Open flame tests .....	3-13
3.2.4.	Inert atmosphere pyrolysis chamber and silica tube .....	3-13
3.2.5.	Thermal analysis.....	3-14
3.2.6.	X-ray diffraction analysis .....	3-14
3.2.7.	Infrared and Raman spectroscopy .....	3-15
3.2.8.	Matrix assisted laser desorption ionisation – time of flight – mass spectrometry .....	3-15
3.2.9.	Pyrolysis gas chromatography – mass spectrometry .....	3-15
3.2.10.	Thermal conductivity measurements.....	3-16
3.2.11.	Electric conductivity measurements.....	3-16
3.2.12.	Electron microscopy.....	3-16
3.2.13.	Surface area measurements .....	3-17
3.2.14.	Burn tests.....	3-17
3.2.15.	Growing single crystals of calcium gluconate monohydrate .....	3-19
3.3.	Method.....	3-20
3.3.1.	Preparation and study of the organic acids' sodium salts .....	3-20
3.3.2.	Preparation and studies of the metal complexes of acetylacetone and gluconic acid .....	3-22
3.3.2.1.	Acetylacetone metal complexes .....	3-22
3.3.2.2.	Gluconic acid metal salts .....	3-23
3.3.3.	Characterisation and evaluation of the intumescence calcium gluconate monohydrate and ammonium gluconate hydrate.....	3-24
3.3.3.1.	The characterisation of the intumescent foams.....	3-26
3.3.3.2.	Thermal analysis of the gluconates.....	3-27
3.3.3.3.	Identification of the gaseous decomposition products .....	3-27
3.3.3.4.	Identification of the solid decomposition products.....	3-28
3.3.3.5.	Molecular mass determination of the carbon foam from the calcium gluconate .....	3-28
3.3.3.6.	Density calculations of the calcium gluconate monohydrate foam .....	3-29
3.3.3.7.	Thermal conductivity of the intumesced calcium gluconate .....	3-30
3.3.3.8.	Electric conductivity measurements .....	3-30
3.3.3.9.	Paint preparation and burn tests.....	3-31
3.3.3.10.	Growing single crystals of calcium gluconate monohydrate .....	3-32
3.3.4.	Preparation of the calcium and other metal dextrose derivatives and study of their intumescence .....	3-34

---

3.3.5. Characterisation of the metal carbonates and oxides.....	3-35
3.3.6. Understanding metal catalysed intumescence of polyols .....	3-35
<b>4. Results and Discussion .....</b>	<b>4-1</b>
4.1. The organic acid sodium salts .....	4-1
4.2. The metal complexes of acetylacetone and gluconic acid.....	4-5
4.3. The intumescence of the calcium and ammonium gluconates .....	4-15
4.3.1. The intumescent foam .....	4-15
4.3.2. Thermal analysis.....	4-22
4.3.3. The gaseous decomposition products .....	4-25
4.3.4. The solid decomposition products.....	4-27
4.3.4.1. XRD analysis.....	4-27
4.3.4.2. IR spectroscopy .....	4-32
4.3.5. The molecular mass of the carbon residue .....	4-35
4.3.6. Foam density as a function of pyrolysis temperature and time.....	4-36
4.3.7. The thermal conductivity of the foam .....	4-38
4.3.8. The electric conductivity of the foam.....	4-39
4.3.9. The fire retardancy of calcium gluconate monohydrate coatings .....	4-42
4.3.10. The crystal data of calcium gluconate monohydrate .....	4-43
4.4. The intumescence of the metal glucose (dextrose) derivatives .....	4-46
4.5. Properties of the fine powder carbonates and oxides .....	4-47
4.6. Understanding metal catalysed intumescence of polyols .....	4-50
<b>5. Conclusion and recommendations .....</b>	<b>5-1</b>
<b>6. References .....</b>	<b>6-1</b>
<b>7. Appendices.....</b>	<b>7-1</b>
7.1. Appendix A .....	7-1
7.1.1. Processing temperatures for commercial polymers .....	7-1
7.2. Appendix B.....	7-2
7.2.1. Limiting Oxygen Index for commercial polymers .....	7-2
7.3. Appendix C.....	7-3
7.3.1. List and structure of acids (and complexes) used .....	7-3
7.4. Appendix D .....	7-8
7.4.1. Acetylacetone complexes used.....	7-8
7.5. Appendix E.....	7-9

---

7.5.1. The commercial preparation of gluconic acid and its derivatives .....	7-9
7.6. Appendix F .....	7-11
7.6.1. Vitamin supplement label.....	7-11
7.7. Appendix G .....	7-12
7.7.1. Pictures of the burn test setup.....	7-12
7.8. Appendix H .....	7-13
7.8.1. Screen grab of the data logging software, “Capture” .....	7-13
7.9. Appendix I.....	7-14
7.9.1. Photos of the cold finger used for the sublimation crystallisation.....	7-14
7.10. Appendix J.....	7-15
7.10.1. Elemental analysis of the leached SiO <sub>2</sub> from Foskor Pty. Ltd.....	7-15
7.11. Appendix K .....	7-16
7.11.1. Preparation of CaDex (Venter, 2000).....	7-16
7.12. Appendix L.....	7-18
7.12.1. Tabulated results for the pyrolysis of the sodium compounds and the synthesis and pyrolysis of the sodium salts .....	7-18
7.13. Appendix M.....	7-31
7.13.1. Summarised results for the gluconate synthesis.....	7-31
7.13.2. Thermal analysis of pentaerythritol, the acetylacetones and acetylacetone/pentaerythritol mixtures.....	7-31
7.13.3. Thermal analysis of the gluconates.....	7-42
7.14. Appendix N .....	7-47
7.14.1. SEM images of calcium gluconate monohydrate powder (crystals).....	7-47
7.14.2. SEM images of ammonium gluconate hydrate (crystals).....	7-48
7.14.3. SEM images of the plate like leached SiO <sub>2</sub> .....	7-48
7.14.4. SEM images of calcium gluconate pyrolysed in air at selected temperatures .....	7-49
7.14.5. SEM images of calcium gluconate monohydrate pyrolysed in nitrogen at selected temperatures .....	7-57
7.14.6. SEM images of calcium gluconate monohydrate and leached silica mixtures pyrolysed in air.....	7-61
7.14.7. SEM images of ammonium gluconate hydrate pyrolysed in air at selected temperatures .....	7-63
7.14.8. SEM images of ammonium gluconate hydrate pyrolysed in nitrogen at selected temperatures .....	7-65
7.14.9. SEM images of AP750 pyrolysed in air at 400 °C .....	7-68
7.14.10. SEM images of PEN pyrolysed in air at 400 °C.....	7-69

---

7.15. Appendix O .....	7-70
7.15.1. Thermal decomposition analysis of gluconic acid.....	7-70
7.15.2. Thermal decomposition analysis of calcium gluconate monohydrate .....	7-71
7.15.3. Thermal decomposition analysis of ammonium gluconate hydrate.....	7-73
7.15.4. Thermal decomposition analysis of the leached silica (ex Foskor) .....	7-75
7.15.5. Thermal decomposition analysis of the expandable graphite (ex Fedmis) .....	7-75
7.16. Appendix P .....	7-76
7.16.1. XRD pattern of calcium gluconate monohydrate .....	7-76
7.16.2. XRD pattern of ammonium gluconate hydrate.....	7-76
7.16.3. XRD pattern of Leached silica from Foskor Pty. Ltd.....	7-77
7.16.4. XRD pattern of calcium gluconate monohydrate pyrolysed in air .....	7-78
7.16.5. XRD pattern of calcium gluconate pyrolysed in nitrogen .....	7-86
7.16.6. XRD pattern of calcium gluconate monohydrate – leached silica mixtures pyrolysed in air.....	7-90
7.16.7. XRD pattern of ammonium gluconate hydrate pyrolysed in air .....	7-94
7.16.8. XRD pattern of ammonium gluconate pyrolysed in nitrogen.....	7-98
7.16.9. IR spectra of calcium gluconate monohydrate pyrolysed in air .....	7-101
7.16.10. IR spectra of calcium gluconate pyrolysed in nitrogen .....	7-101
7.16.11. IR spectra of ammonium gluconate hydrate pyrolysed in air .....	7-102
7.16.12. IR spectra of ammonium gluconate pyrolysed in nitrogen.....	7-102
7.16.13. Decomposition products of calcium gluconate monohydrate.....	7-103
7.16.14. Thermal conductivity results from the SABS.....	7-104
7.17. Appendix R .....	7-106
7.17.1. Electric conductivity for the pyrolysed ammonium gluconate .....	7-106
7.18. Appendix S .....	7-107
7.18.1. Burn through tests for the painted balsa wood planks – Graphs.....	7-107
7.18.2. Burn through tests for the painted balsa wood planks – Pictures .....	7-111
7.18.3. Burn through tests for the painted aluminium plates – Graphs.....	7-113
7.18.4. Burn through tests for the painted aluminium plates – Pictures .....	7-115
7.18.5. Burn through tests for the painted cardboard sheets – Graphs .....	7-116
7.19. Appendix T .....	7-118
7.19.1. Light microscope and SEM images of calcium gluconate monohydrate crystals (powder) .....	7-118
7.19.2. Light microscope and SEM images of calcium gluconate monohydrate crystals recrystallised through diffusion technique.....	7-119
7.20. Appendix U .....	7-121
7.20.1. Metal oxides and carbonates prepared from the metal dextrose solutions and calcium gluconate monohydrate .....	7-121

---

7.21. Appendix V .....	7-127
7.21.1. Thermal analysis of selected calcium salts .....	7-127
7.21.2. SEM images of glyceric acid hemicalcium salt monohydrate at selected temperatures in air .....	7-131

## LIST OF FIGURES

Figure 2-1	Emman's fire triangle.....	2-2
Figure 2-2	A simplified model for combustion and flame retardancy.....	2-3
Figure 2-3	The property (P) evaluation of a two component system blend.....	2-7
Figure 2-4	Schematic diagram for acid catalysed intumescence .....	2-14
Figure 2-5	Acid catalysed dehydration of a hydroxyl compound.....	2-16
Figure 2-6	The action of the phosphoric acid in char formation mechanism .....	2-16
Figure 2-7	A single intumescent complex containing all three functionalities .....	2-17
Figure 3-1	2,4-Pentanedione (acetylacetone) and its stable isomers .....	3-4
Figure 3-2	The structures of D-glucose and D-gluconic acid .....	3-6
Figure 3-3	The structure of calcium D-gluconate monohydrate .....	3-7
Figure 3-4	The structure of ammonium D-gluconate hydrate .....	3-7
Figure 3-5	Schematic representation of saccharinic acids .....	3-10
Figure 3-6	Glass tube for the pyrolysis of the samples and a tube in the aluminium block on the furnace door.....	3-13
Figure 3-7	Glass chamber used for the pyrolysis in an inert atmosphere .....	3-13
Figure 3-8	Assembly housing the five thermocouples.....	3-18
Figure 3-9	Side view of the burn test setup .....	3-18
Figure 3-10	Back view of the burn test setup showing the thermocouples.....	3-19
Figure 3-11	Setup for the sublimation crystallisation of calcium gluconate.....	3-20
Figure 3-12	An empty glass tube (left) and tubes with pyrolysed samples (right) of the sodium salts .....	3-21
Figure 3-13	Diffusion crystallisation of calcium gluconate monohydrate.....	3-33
Figure 3-14	Structures of the compounds studied in this section .....	3-36
Figure 4-1	TGA/DSC curves of a 1:1 mass ratio copper (II) acetylacetone and pentaerythritol mixture obtained at a scan rate of 10°C per minute in air .....	4-7
Figure 4-2	TGA/DSC curves of a 1:1 mass ratio calcium acetylacetone and pentaerythritol mixture obtained at a scan rate of 10°C per minute in air .....	4-7
Figure 4-3	Comparison of the acetylacetone/pentaerythritol mixtures' TGA curves .....	4-8
Figure 4-4	Comparison of the acetylacetone/pentaerythritol mixtures' DSC curves.....	4-9

---

Figure 4-5	TGA/DSC curves of copper gluconate obtained at a scan rate of 10°C per minute in air .....	4-10
Figure 4-6	TGA/DSC curves of magnesium gluconate obtained at a scan rate of 10°C per minute in air .....	4-10
Figure 4-7	TGA/DSC curves of sodium gluconate obtained at a scan rate of 10°C per minute in air .....	4-11
Figure 4-8	TGA/DSC curves of gluconic acid obtained at a scan rate of 10°C per minute in air .....	4-12
Figure 4-9	Comparison of the gluconic acid salts' TGA curves.....	4-13
Figure 4-10	Comparison of the gluconic acid salts' DSC curves .....	4-13
Figure 4-11	A small sample (~1 g) of calcium gluconate monohydrate in a polytop® before pyrolysis in a laboratory furnace at 300°C for 5 minutes .....	4-16
Figure 4-12	The calcium gluconate sample being foamed in the furnace at 300°C.....	4-16
Figure 4-13	The same calcium gluconate sample after being heated at 300°C for 5 minutes .....	4-17
Figure 4-14	SEM images of the foam structure of the oven-pyrolysed sample obtained at 400°C in air.....	4-20
Figure 4-15	AP750 pyrolysed at 400°C in air for 5 minutes .....	4-21
Figure 4-16	PEN pyrolysed at 400°C in air for 5 minutes .....	4-22
Figure 4-17	SDTA/TGA curves of 50% gluconic acid solution in water obtained at a scan rate of 10°C per minute in air .....	4-23
Figure 4-18	TGA/DSC curves of calcium gluconate obtained at a scan rate of 10°C per minute in air .....	4-24
Figure 4-19	TGA/DSC curves of calcium gluconate monohydrate obtained at a scan rate of 10°C per minute in an inert (argon) atmosphere .....	4-24
Figure 4-20	Thermal decomposition of calcium acetate.....	4-26
Figure 4-21	Calcium gluconate monohydrate heated at 150°C for 5 minutes (air) .....	4-29
Figure 4-22	Calcium gluconate monohydrate heated at 600°C for 5 minutes (air) .....	4-29
Figure 4-23	Calcium gluconate monohydrate heated at 750°C for 5 minutes (air) .....	4-30
Figure 4-24	Calcium gluconate monohydrate heated at 1000°C for 5 minutes (air) .....	4-30
Figure 4-25	Calcium gluconate monohydrate heated at 600°C (top) and 700°C (bottom) for 5 minutes (inert).....	4-31
Figure 4-26	Comparison of pyrolysis in air and inert atmospheres .....	4-33
Figure 4-27	Comparison of calcium gluconate monohydrate and samples pyrolysed at low temperatures in air .....	4-33
Figure 4-28	Comparison of samples heated at temperatures of 700°C and above in air .....	4-34

---

Figure 4-29	MALDI-TOF-MS spectrum of calcium gluconate monohydrate pyrolysed at 300°C for 5 min in air .....	4-36
Figure 4-30	Mass loss and density of calcium gluconate foam as a function of temperature, pyrolysed for 5 minutes in air .....	4-37
Figure 4-31	Mass loss and density of calcium gluconate foam as a function of pyrolysis time at 300°C in air .....	4-38
Figure 4-32	Electric conductivity measurements for embedded graphite, coked calcium and ammonium gluconate at 1, 2 and 5 volt.....	4-40
Figure 4-33	Conductivity measurements for embedded coked ammonium gluconate .....	4-41
Figure 4-34	Light microscope images of commercial calcium gluconate monohydrate crystals (powder).....	4-43
Figure 4-35	SEM images of calcium gluconate monohydrate crystals (powder) .....	4-44
Figure 4-36	Light microscope images of calcium gluconate monohydrate recrystallised from water through diffusion with ethanol.....	4-45
Figure 4-37	SEM images of calcium gluconate monohydrate recrystallised from water through diffusion with ethanol .....	4-45
Figure 4-38	Aluminium oxide prepared from aluminium ammonium dextrose solution .....	4-48
Figure 4-39	Calcium carbonate prepared from calcium dextrose solution .....	4-48
Figure 4-40	Calcium carbonate prepared from calcium gluconate monohydrate .....	4-48
Figure 4-41	Iron oxide prepared form iron dextrose solution.....	4-49
Figure 4-42	Densities of several hydroxyl rich and dehydrated compounds pyrolysed at selected temperatures in air .....	4-52

## LIST OF TABLES

Table 2-1	Estimated flame retardant consumption for 1990 .....	2-6
Table 3-1	Typical properties for CaCO <sub>3</sub> paint extenders.....	3-12
Table 3-2	XRD instrument and data collection parameters.....	3-14
Table 3-3	Temperatures selected pyrolysis for the calcium gluconate monohydrate.....	3-25
Table 3-4	Temperatures selected pyrolysis for the ammonium gluconate hydrate .....	3-25
Table 3-5	Amounts of leached silica used .....	3-26
Table 4-1	The effect of organic structures on carbon formation .....	4-2
Table 4-2	Response of powder samples to the application of an open LPG flame .....	4-6
Table 4-3	Comparison of the onset and char oxidation temperatures and oxidation exotherm peak width for the metal complexes from the thermal analysis .....	4-14

---

Table 4-4	Comparison of the carbon char yield at selected temperatures for all the gluconates and acetylacetone/pentaerythritol mixtures.....	4-15
Table 4-5	Decomposition products of calcium gluconate monohydrate (air) .....	4-28
Table 4-6	Decomposition products of calcium gluconate monohydrate (inert) .....	4-31
Table 4-7	Summary of the molecular masses obtained from the MALDI-TOF-MS.....	4-36
Table 4-8	Thermal conductivity (at 300 K) and density of insulating materials .....	4-39
Table 4-9	Results form the burn-through tests of the coated balsa wood planks, aluminium plates and cardboard sheets.....	4-42
Table 4-10	BET surface area results from the metal oxides and carbonates prepared from the metal dextrose solutions .....	4-47
Table 4-11	Elemental analysis of the metal dextrose solutions as obtained from the EDS SEM analysis.....	4-49
Table 4-12	Density data of several hydroxyl rich and dehydrated compounds pyrolysed at selected temperatures in air .....	4-51
Table 4-13	Summary of the thermal analysis data for the selected compounds.....	4-53
Table 4-14	Theoretical percentage accumulative loss and percentage contribution of selected species .....	4-53

## NOMENCLATURE

Symbol	Property	Unit	
A	cross section area	square meter	[m <sup>2</sup> ]
I	electric current	Ampere	[A]
<i>l</i>	electric path length	meter	[m]
R	electrical resistance	Ohm	[Ω]
V	electric potential	volt	[V]
Σ	electric conductivity	-	[Ω <sup>-1</sup> .m <sup>-1</sup> ]

## ABBREVIATIONS

ABS	acrylonitrile-butadiene-styrene
ASTM	American Society for Testing of Materials
APP	ammonium polyphosphate
ATH	aluminium trihydroxide or alumina trihydrate
BET	Brunauer, Emmett and Teller

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ca.	<i>circa</i> (approximately)
CaDex	calcium complexes produced from dextrose monohydrate
DMSO	dimethyl sulfoxide
DSC	differential scanning calorimetry
DTA	differential thermal analysis
e.g.	<i>exempli gratia</i> (for example)
EDS	energy dispersive spectroscopy
EDX	energy dispersive analysis of X-rays
<i>et al.</i>	<i>et alibi</i> (and elsewhere)
etc.	<i>et cetera</i> (and so forth)
FeDex	iron complexes produced from dextrose monohydrate
FEP	perfluoro(ethylene/propylene)
IR	infrared
LCP	liquid crystal polymers
LOI	limiting oxidation index
LPG	liquefied petroleum gas
LV	low vacuum
MALDI-TOF-MS	matrix assisted laser desorption ionisation – time of flight – mass spectrometry
MetDex	metal complexes produced from dextrose monohydrate
P	property
PA	polyamide
PAI	poly(amide imide)
PAN	polyacrylonitrile
PBT	poly(butylene terephthalate)
PC	polycarbonate
PE	polyethylene
PEEK	poly(ether ether ketone)
PEI	poly(ester imide)
PEO	poly(ethylene oxide)
PER	pentaerythritol
PES	polyethersulphone
PET	poly(ethylene terephthalate)
PMMA	poly(methyl methacrylate)

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POM	polyoxymethylene
PP	polypropylene
PPO	poly(phenylene oxides)
PPS	poly(phenylene sulphide)
PS	polystyrene
PSU	polysulphone
PTFE	poly(tetrafluoroethylene)
PUR	polyurethane
PVA	poly(vinyl alcohol)
PVC	poly(vinyl chloride)
rpm	revolutions per minute
SABS	South African Bureau of Standards
SAN	styrene-acrylonitrile
SDTA	simultaneous differential thermal analysis
SEM	scanning electron microscope
TEM	transmission electron microscope
TGA	thermo gravimetric analysis
vs.	<i>versus</i> (against)
XRD	X-ray diffraction